

SIMULACIÓN POR AUTÓMATAS CELULARES DE LAS CONFORMACIONES ESPACIALES DE POLIELECTROLITOS.

Carlos Echeverría^{1,2*}, Wilmer Olivares-Rivas², and Kay Tucci³

¹*Lab. de Física Aplicada y Computacional, Depto. de Matemática y Física, Universidad Nacional Experimental del Táchira, Táchira, Venezuela.*

²*Grupo de Quimicofísica de Fluidos y Fenómenos Interfaciales, Depto. de Química, Fac. de Ciencias, Universidad de Los Andes, Mérida, Venezuela.*

³*SUMA-CeSiMo, Fac. de Ciencias, Universidad de Los Andes Mérida, Venezuela.*

RESUMEN

Hemos llevado a cabo simulaciones del tipo Autómata Celular para un modelo de un polielectrolito en dilución infinita, para reproducir de manera cualitativa sus propiedades conformacionales. Nuestros resultados predicen las llamadas estructuras de *collar de perlas*, las cuales se comparan bien con simulaciones de Dinámica Molecular más elaboradas y costosas.

CELLULAR AUTOMATA SIMULATION OF THE SPATIAL CONFORMATIONS OF POLYELECTROLYTES.

ABSTRACT

We carried out a Cellular Automata simulation of a model polyelectrolyte solution at infinite dilution, in order to reproduce qualitatively its conformational properties. Our results predict the so called *pearl necklace* structures, which compare favorably with the more elaborated and costly Molecular Dynamics simulations.

PACS numbers:

I. INTRODUCTION

Polyelectrolyte (PE) solutions are systems widely studied since they show properties that are of fundamental interest for applications in health science, food industry, water treatment, surface coatings, oil industry, among other fields. In fact, one of the problems found in genetic engineering is the appearance of conformational changes of the ADN molecule, which is a charged polyelectrolyte.[3].

Here we study an infinite dilution polyelectrolyte solution, so that, the interaction among polyelectrolyte

macromolecules are negligible. We model the polyelectrolyte as having dissociable functional groups that give rise to charged sites and counter-ions in aqueous solution. The long range interactions arising from these multiple charges are responsible for their macroscopic complex properties, which can not be explained by regular polymer theories. The spatial structures of these materials in solution have been studied extensively, particularly with a scaling theory[1, 2] that are not appropriate for highly charged PE. The first simulations carried out for a single chain predicted the formation of groups of monomers, as the fraction of charged monomers increased. Such structures are known as pearl necklaces. The size of such pearls and the distance between them is determined by the balance between the electrostatic repulsion and steric effects.

*Corresponding author: e-mail: cecheve@ula.ve

These pearl necklace structures have also been found in Molecular Dynamics (MD) simulations[4, 5, 6, 7, 8, 9, 10]. In this paper we are interested in the application of the much simpler Cellular Automata simulation to characterize the main features of a polyelectrolyte that could be responsible for such conformations. The complete simulation of this complex system requires the description of a model in terms of potential or forces. In the MD simulations of Limbach and Holm [4], the monomers are connected along the chain by the finite extendible non-linear elastic (FENE) bond represented by the potential energy.

$$U_E(r) = -\frac{1}{2}K_ER_0^2 \ln \left(1 - \frac{r^2}{R_0^2} \right) \quad (1)$$

where r is the distance between two bonded monomers, K_E is the elastic bond constant, σ is the monomer diameter, k_B is Boltzmann's constant, T is the absolute temperature and the parameter R_0 represents the maximum extension of the bond between two neighbor monomers. Two charged sites i and j , with charges eq_i and eq_j , a distance r_{ij} apart, interact with the electrostatic Coulomb potential

$$U_C(r_{ij}) = k_BT \frac{\lambda_B q_i q_j}{r_{ij}}. \quad (2)$$

This potential is weighed by the Bjerrum length $\lambda_B = e^2/[4\pi\epsilon_o\epsilon_s k_BT]$, where ϵ_s and ϵ_o are the solvent permittivity and the vacuum permittivity respectively, e is the electric charge unit. The parameter λ_B is a measure of the strength of the electrostatic force as compared to the kinetic energy. The length ratio σ/λ_B is a measure of the reduced temperature $T/[e^2/(4\pi\epsilon_o\epsilon_s k_B\sigma)]$.

The short range and van der Waals interaction between any two particles or monomers is represented in the MD simulation by a typical truncated Lennard-Jones potential

$$U_{LJ}(r_{ij}) = \begin{cases} 4\epsilon \left[\left(\frac{\sigma}{r_{ij}} \right)^{12} - \left(\frac{\sigma}{r_{ij}} \right)^6 \right] + \epsilon_{cut} & r_{ij} < R_c, \\ 0 & r_{ij} > R_c \end{cases} \quad (3)$$

where ϵ is the potential energy well depth and ϵ_{cut} is the cut off energy. This potential prevents the superposition of the bonding monomers. Counter-ions interact via a purely repulsive LJ interaction with $R_c = 2^{1/6}\sigma$.

II. CELLULAR AUTOMATA MODEL

Even though in the Cellular Automata simulation we do not use any form of potential energies or forces in an explicit manner, the *rules* for the movement of the different particles must be inspired on a model defined in terms of such potentials. We therefore establish our rules based on the essence of the previous three

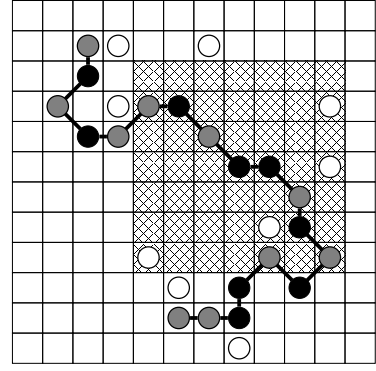


FIG. 1: *Cellular Automata Model.* The dark bonded circles represent the charged monomers, the gray circles are the neutral monomers and the open circles are the counter-ions. The shaded area corresponds to the range volume $\lambda_B = 3\sigma$

potentials.

A. Polyelectrolyte Chain Construction Rules

The polymer is constructed by placing the monomers in a three dimensional cubic network of side L and volume L^3 . Each cell then has 26 neighbors and represents a monomer with a monovalent charge $+1$, -1 , or, for a neutral monomer, 0 , as depicted in Fig.(1) in two dimensions. Out of a total of N monomers in the chain, it is assumed that a given fraction f is charged. The polymer is then constructed by randomly binding consecutive sites in the network. Each monomer could be charged or uncharged, with a distribution chosen randomly. A key step on the construction of the polyion is the spatial location of the dissociated counter-ions. We place the counter-ions also randomly in free cells in the volume around the charged monomer within a distance λ_B , that is, in a volume $(2\lambda_B)^3$ centered on the charged site. The use of the Bjerrum parameter, which is related to the quality of the solvent, ensures the conservation of the total electroneutrality but gives a spatial distribution of counter-ions around the charged sites.

So, each monomer i of the system is represented in a $N \times 4$ matrix where each element $M(m_x, m_y, m_z, q_i)$ indicates the i^{th} polymer with charge q_i , that could be -1 , $+1$ or 0 , at the positions x, y, z given by the cell label $[m_x, m_y, m_z]$. The counter-ions with opposite charge are represented by a similar $fN \times 4$ matrix M_C .

For simplicity we chose monomers with dissociable groups that give a site with a positive charge. We then set the following displacement rules for the different

particles:

B. Displacement Rules

Neutral Monomer Particle

1. Locate the unoccupied nearest neighbor sites. The new position where a move could be acceptable are those where it does not superimposed with any other particle in the system and where no bond is broken.
2. Count the amount of monomer particles around the current position and around every unoccupied neighbor, within a cube of volume $(2L)^3$ centered on it.
3. Move the test neutral monomer to the position that has the higher amount of monomers around it, including the current one if it were the case.

Positively Charged Monomer Particle

1. As before, locate the unoccupied acceptable nearest neighbor sites.
2. Count the amount of charge around the current position and around every unoccupied neighbor, within a cube of volume $(2\lambda_B)^3$ centered on it.
3. Move the test positive charged monomer to the position that has the lowest positive charge around it, including the current one if it were the case.

Negatively charged counter-ion

1. Move randomly to an unoccupied site within a cube of volume $(2\lambda_B)^3$ centered on the accepted new position of the corresponding positive monomer in the polymeric chain.

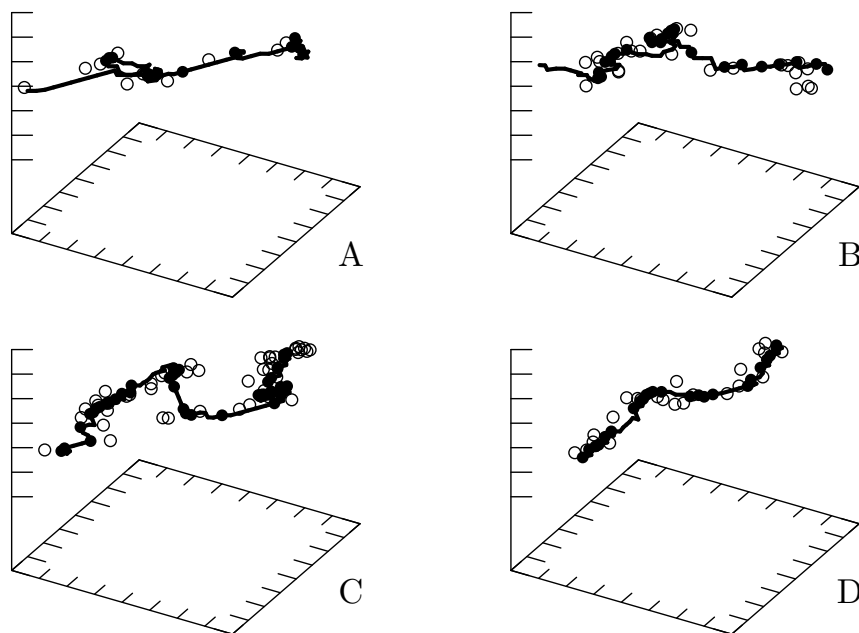


FIG. 2: Final polyelectrolyte configuration for different charge fractions f : A) 0.1 B) 0.3 C) 0.5 D) 0.7. The pearl necklace conformations appear as the charge fraction increases.

III. RESULTS AND DISCUSSION

We denote the position of monomer i with r_i and the distance between two particles i and j with r_{ij} . The center of mass for the chain is then $R_s = \frac{1}{N} \sum_{i=1}^N r_i$.

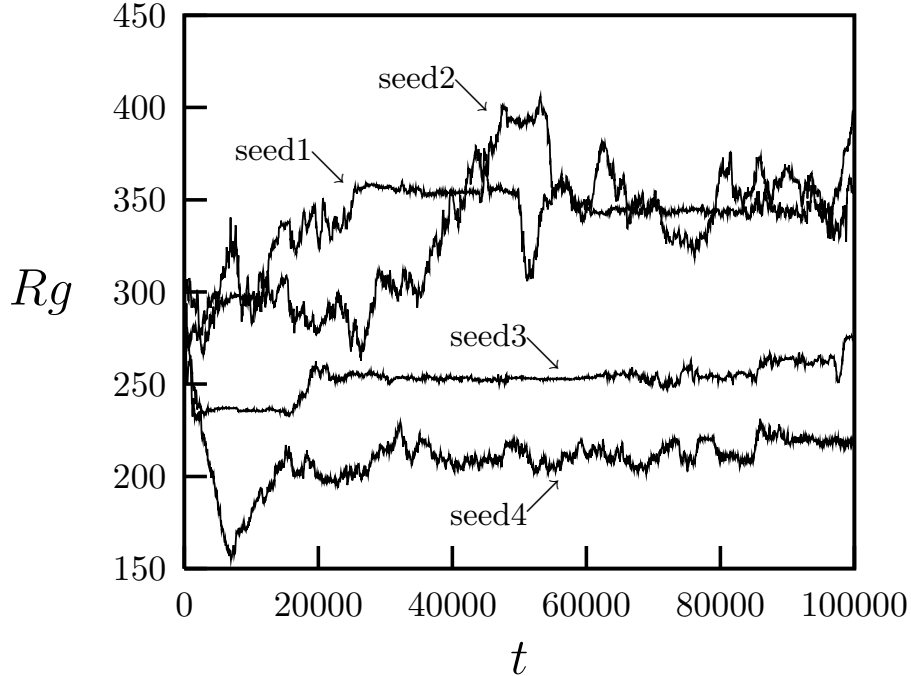


FIG. 3: Evolution of the Radius of Gyration R_g as a function of time steps t , for $N = 100$, $\lambda_B/\sigma = 3$ and $f = 0.5$. Each curve corresponds to different random charged sites distributions, obtained with different seeds.

and the center of mass coordinates are $x_i = r_i - R_s$. A parameter that is useful in the study of the spatial conformations of a polymer is the radius of gyration R_G , defined as

$$R_g^2 = \frac{1}{N} \sum_{i=1}^N x_i^2 \quad (4)$$

According with our construction and movement rules, we can vary the length of the chain L , or the number of monomers N , satisfying $L = N\sigma$ and the number of charged monomers, $N_c = fN$. We also take as an independent variable the parameter λ_B , which determines the number of cells, of size σ , where the range of the electrostatic attraction between a charged monomer and contra-ion extends. The charge distribution of the sequence of charged and uncharged monomers is determined randomly depending on the initial random seed.

In Fig.(2) we show some of the equilibrium conformations obtained for a polyelectrolyte with $N = 100$ monomers, with $\lambda_B = 3\sigma$, for several charge fractions. The line represents the polyelectrolyte, the filled black bonded circles represent the charged monomers. The counter-ions in solution are represented by open non-bonded circles. For clarity, the neutral monomers are

not shown. For a low charge fraction of $f = 0.1$, the polyion presents an elongated string appearance. As the charge fraction is increased the polyion contracts and some groups of monomers tend to form clusters, so that, already for $f = 0.5$, it shows the locally collapsed structures known as pearl necklace. These results are very similar to those obtained by the Molecular Dynamics simulations of Limbach and Holm [2, 4]

It is important to notice that for a given total charge, determined by the fraction f , different distributions of the charged monomers give different conformations. To study this behavior, we have carried out simulations with several initial seeds for $N = 100$ monomers, with a fixed fraction $f = 0.5$ and a range parameter of $\lambda_B/\sigma = 3$. In Fig.(3), we show the temporal evolution of the radius of gyration R_g . In all cases the conformations change from the initial given R_g value to a plateau value that correspond to the equilibrium structures. Fig.(3) clearly show that the plateau R_g values, and thence, the final conformations depend on the charge distribution. In Fig.(4) we show snapshots of final structures corresponding with the different seeds of Fig.(3).

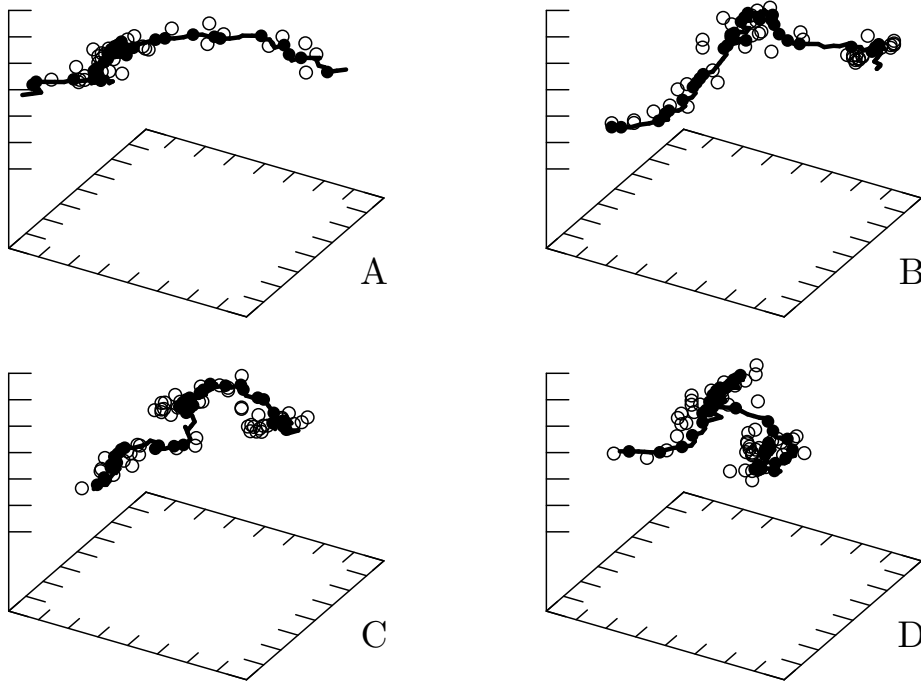


FIG. 4: *Equilibrium Polyelectrolyte structures in the plateau R_g region for the charge of distributions of Fig.(3). A) seed1, B) seed2, C) seed3 and D) seed4.*

As we can see from Fig.(4), the formation of the pearl necklace structures is independent of the charge sites distribution, for a given f and λ_B . These clusters seem to be stabilized by the counter-ions as a consequence of the electroneutrality condition that we force to be satisfied. This is so because in our model the degree of freedom of

the mobile counter-ions is much higher than that of the monomers tied to the chain. The strong repulsions that originate by the formation of clusters of neutral and positively charged monomers is compensated by the counter-ion cloud that forms around it.

In order to study the reproducibility of the configurations found, we carried out a large number of simulation runs, for fixed values of $N = 150$, $f = 0.5$, $\lambda_B/\sigma = 3$ and 9 and for the same initial charge distribution. In Fig.(5) we show the histograms for the frequency with

which a given value of R_g appears. We can see that the distribution of the R_g is very closely a gaussian with a reasonably low dispersion of less than a 10% about the mean value.

In Fig.(5B) we can see that similar distribution is ob-

tained when the range parameter λ_B/σ increases to a

value of 9. We have further tested the effect of the parameter λ_B by generating structures for different values of it. In Fig.(6) we show some equilibrium conformations for λ_B/σ equal to A) 3,B) 6,C) 9 and D) 12. Here we use a large charge fraction $f = 0.5$ and $N = 100$ and we used the same initial charge distributions in all cases. As we can observe as Bjerrum length increases the final polyelectrolyte structures become more compact. The number of pearls or conglomerates is higher for the lower values of λ_B , a result similar to that obtained by MD simulations of Limbach y Holm [4]-[10]

IV. CONCLUSIONS

With the simple technique described here, we were able to reproduce the complex structure of model polyelectrolytes that fare very well with those predicted by the more sophisticated Molecular Dynamic and Monte Carlo simulations[2, 4]. We even predict situations with single

conglomerates and with pearl necklace type conglomerates. We thus show the potentiality of the Celular Automata in the simulation of the trends in the formation of the various types of spatial conformations of polyelectrolytes. We remark on the importance of the charge distribution once the fractional charge is fixed.

V. ACKNOWLEDGEMENT

This work was supported in part by the grant 04-005-01 from the Decanato de Investigación of Universidad Nacional Experimental del Táchira, in part by the grant G-9700741 of FONACYT, and in part by the grant C-1279-0402-B from Consejo de Desarrollo Científico Humanístico y Tecnológico of Universidad de Los Andes. Numerical calculations were carried out at the computer center CeCalcULA.

-
- [1] A.V. Dobrynin, R.H. Colby and M. Rubinstein, *Macromolecules*, **28**,1859,(1995).
 - [2] A. V. Dobrynin; M. Rubinstein; S. P. Obukhov, *Macromolecules*, **29**,2974,(1996).
 - [3] Y. Yamasaki, Y. Teramoto, and K. Yoshikawa, *Biophys. J.*, **80**,2974,(2001).
 - [4] H. J. Limbach y C. Holm, *J. Phys. Chem. B*, **107**,8041-8055,(2003).
 - [5] U. Micka; C. Holm; K. Kremer, *Langmuir*, **15**,4033,(1999).
 - [6] U. Micka; K. Kremer, *Europhys. Lett.*, **49**,189-195,(2000).
 - [7] H. J. Limbach y C. Holm, *J. Chem. Phys.*, **114**,9674-9682,(2002).
 - [8] H. J. Limbach y C. Holm, *Comput. Phys. Commun.*, **147**,321-324,(2002).
 - [9] H. J. Limbach y C. Holm; K. Kremer, *J. Phys: Condens. Matter*, **15**,S205-S211,(2003).
 - [10] H. J. Limbach y C. Holm; K. Kremer, *Europhys. Lett.*, **60**,566-572,(2002).

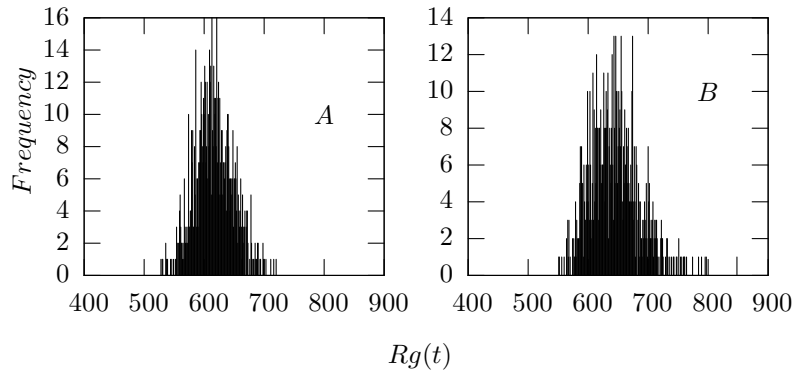


FIG. 5: Histograms for R_g for $N = 150$, $f = 0.5$ and two values of λ_B/σ , A) 3 and B) 9

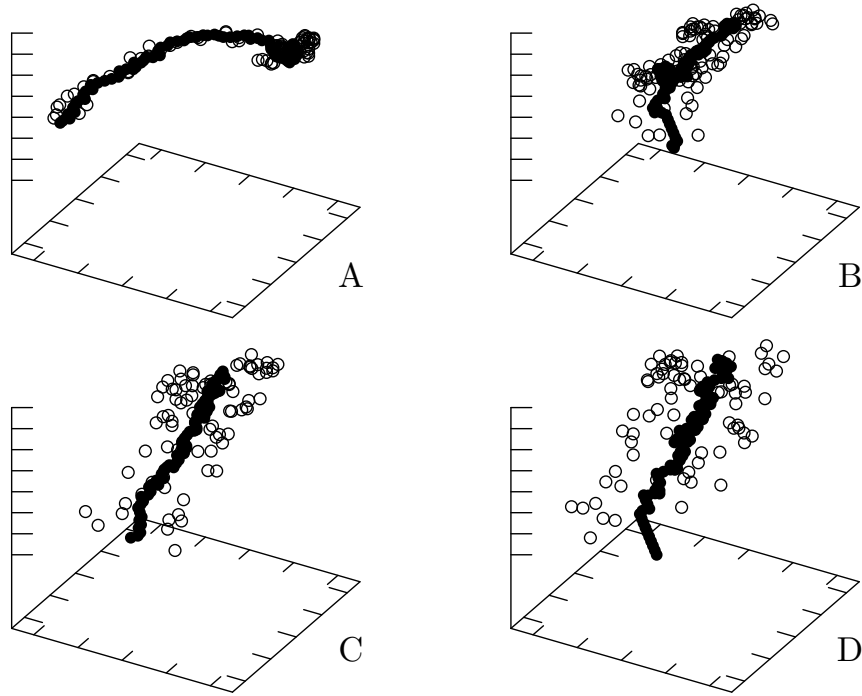


FIG. 6: Polyelectrolyte conformations for several values of the reduced Bjerrum parameter λ_B/σ A) 3, B) 6, C) 9 and D) 12. Here $f = 0.9$ and $N = 100$ and we used the same initial charge distributions